10th International Conference on Composite Science and Technology ICCST/10 A.L. Araújo, J.R. Correia, C.M. Mota Soares, et al. (Editors) © IDMEC 2015

IN-AUTOCLAVE RESIDUAL STRESS PROGRESSION DURING MULTIPLE STEP FABRICATION OF COMPOSITE CYLINDERS

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Key words: Composites, Manufacturing, FBG, Process-induced deformation, Cure Monitoring.

Summary: Process-induced deformation represents a risk factor for composite structures as they affect surely the dimensional stability of the parts or the structural integrity. In this work, residual strains for a thick composite cylinder have been measured by FBG sensors. In addition a numerical procedure for predict strain state at the end of manufacturing process has been developed and compared to experimental data.

1 INTRODUCTION

High performance composite materials are widely used in aerospace industry thanks to their intrinsic benefits such as lightweight, and high stiffness. Composite materials offer the chance to soften exorbitant raw material and manufacturing costs by monolithic design reducing the number of parts and leading to simplified geometries.

Fabrication of high quality thick composite structures is still a major challenge for industry since such structures apt to the built-up of residual stresses related to process issues

The manufacturing process of composite materials is a multi-physical problem since concerns thermodynamics, chemistry and mechanical topics. As matter of facts, thermal conduction within the mould and within the components drives the stress distribution within the part through gradients in degree of cure and/or mechanical distortions.

Quality of components depends on ply deposition and cure process: factor as porosity, dry spots, degradation or cure gradients endanger structural efficiency of the composite parts. In line monitoring techniques allow the enhancement of final features of products by interconnecting raw materials, process and structural investigation [1].

In line sensing technology able to detect thermo-mechanical properties during the composite cure are based on ultrasonic transducers, thermometers, pressure and deformation sensors [2]. Typically, monitoring the cure process within the industry is limited to time and temperature. Usually the manufactures define cure profile according to resin manufacturer instructions. A cure monitoring system [3] offers several benefits such as

- Allowing tailored cure schedules to suit the resins age and chemical integrity,
- Measuring residual stresses,
- Indicating when cure is complete,
- Assisting in maintaining product quality, repeatability and reducing scrap rates by

offering quantitative feedback on the cure process.

Fibre optic sensors revealed their effectiveness for composite application since their dimensions are suitable for embedding within the stacking sequence without alter the composite integrity [2,3,4]. In addition, chemical transformation of the matrix during cure process does not affect the sensing capability of fibre optic sensors.

Karalekas et al. [5] demonstrated the capability of FBG sensors in providing information on the degree of cure and evolution of fabrication induced strains inside the epoxy. In their work a specimen was subjected to different temperature cycles, each cycle consisting of three stages: (a) the ramp-up stage where the temperature was increased to the selected one within 2 h, (b) a plateau where the temperature was kept constant for 9 h, (c) the cooling down stage to room temperature after opening the oven door at the end of the plateau. The cooling stage was fast enough to generate sufficient residual strains in each case.

Thermal deformation and/or external strain influence the spacing between reflective surfaces and their refractive indexes. Environmental strains or temperature induce on optical fibre interferometers a shift in the reflected wavelength peak is experienced.

An FBG is a periodic and permanent modification of the core refractive index value along the optical fibre axis. An FBG is defined by several physical parameters: the grating length, i.e. the gauge length where the index modulation have been realised; the periodicity (Λ) and the amplitude (δ_n). The perturbation in the fibre core induces light coupling in two propagating modes inside the core, the mode coupling occurs for some wavelength around the Bragg wavelength defined by:

$$\lambda_{Bragg} = 2 n_{eff} \Lambda \tag{1}$$

Where n_{eff} is the effective refractive index of the core at Bragg wavelength.

The effective refractive index of the core and spatial periodicity of the grating are both affected by the changes in temperature and strain.

$$\frac{d\lambda}{\lambda} = (1 - P_e) d\varepsilon + (\alpha_{glass} + \zeta) dT$$
⁽²⁾

Where P_e is the elasto-optic coefficient (0.22 adimensional) while the thermal sensitivity is about 6.67 10-6 /°C for optical fibre whose core is silicon dioxide. Actually, because FBGs are made from different fabrication techniques, they show different sensing properties; therefore the strain and temperature sensitivity coefficients of FBG should be determined by calibration before we use it as strain or temperature sensors in practical applications, especially for the packaged FBGs.

In this research, thermo-mechanical deformation achieved during the overall cure of a thick cylinder made by autoclave process has been experimentally investigated by embedding FBG (Fibre Bragg Grating) sensors into the composite. Actual practise for the manufacturing of thick composite cylinders consists of multiple manufacturing steps: at each step, the composite thickness is increased by wrapping additional un-cured layers.

FBG sensors embedded at each step allowed the estimation of temporal variation of strains during the entire fabrication at different positions through the cylinder thickness.

Final results show a relevant strain relaxation at the cure dwell onset whereas residual strain dramatically increases when further layered materials is wrapped

A numerical model of the residual stress built up phenomenon is also presented. The computational analysis accounts for the cure-dependent viscoelastic properties of the transversely isotropic materials for each wrapping steps during the whole manufacturing

process. Issues related to the sensing technology, data analysis and comparison with numerical predictions are presented and analysed.

2 FIBER OPTIC SENSORS INTEGRATION

3.1 Fiber Bragg Grating. Signal processing and temperature compensation

Fiber bragg gratings are sensitive to both temperature and deformation field, the integration of the equation (2) lead to the relation between peak shift and thermo-mechanical forces acting on the sensor:

$$\ln\left(\frac{\lambda}{\lambda_0}\right) = (1 - P_e)\left(\varepsilon - \varepsilon_0\right) + (\alpha_{glass} + \zeta)\Delta T$$
(3)

The deformation at initial state is assumed zero ($\varepsilon_0 = 0$). This condition is correct only at first polymerization step, i.e. when the sensor is embedded on the sample, in further steps the deformation applied on the sensor should consider all previous steps.

Based on latter consideration, λ_0 has been considered for each sensor as nominal value from sensor calibration in temperature calibration test at 30°C. If prestrain is present on the sensor at first installation, the initial strain have been zeroed.

According to equation (3) the strain measured in the sensor is evaluated as:

$$\varepsilon(t) = \frac{10^{6}}{(1-P_{e})} \left[\ln \left(\frac{\lambda(t)}{\lambda_{0}} \right) + k_{T} \left(T(t) - 30^{\circ}C \right) \right] - \varepsilon(t=0)$$
(4)

3.2 Integration of deformation sensors during manufacturing process

The manufacturing process of the test case is divided in five separated steps, at each step three FBG array have been integrated within the composite: two bare array have been embedded directly in the composite material while a third array is considered for temperature compensation during the test.

Figure 1 reports the installation of FBG arrays on the composite cylinder: a previously cut composite layer is first wrapped on the tool (first step) or on the external surface (further steps) of the composites. Finally the plies have been wrapped on the tool with a pneumatic press.



Figure 1: Composite tape and FBG array positioning

FBG Arrays contain two sensors spaced each other thirty centimetres. During the installation the first sensor have been placed thirty centimetres inside the cylinder in order to reduce boundary effect. The installation of free strain sensor is divided in two parts, during wrapping stage an hollow carbon pipe is embedded in the composite cylinder then before bagging stage TS array is inserted. Since carbon pipe has an external diameter of about 700 μ m, a small strip of the UD tape has been cut out from the bottom layer. The temperature FBG array is inserted within the cylinder at the end of manufacturing stage.

When all the required tapes have been wrapped on the tool, the cylinder is moved to a lathe modified to automatically envelop the composite with nylon strips. Below the bag, a fabric is applied in order to make easier the adhesion to further lamination step tapes.

Once the bag have been placed, all optical fibres are arranged on the tool to avoid pull out due to the hot flow inside the autoclave chamber. Optical fibres should not tight on tool to avoid, during its thermal expansion, dangerous loads, which could blind the measure. After optical fibre disposition is complete and that all arrays are completed the cylinder is loaded in the autoclave chamber. Optical fibres mounted at each step have been preserved in order to measure the rise of strains in further manufacturing stages.

Figure 2 reports the sensorization scheme: at each manufacturing step a couple of sensor is added, moreover a temperature sensor is provided for temperature compensation purpose. For each manufacturing step an FBG array has been arranged along 45° fibres and another has been arranged along cylinder axis fibres. Unfortunately the FBG2 (0°) array was lost during manufacturing.



Figure 2: Installation scheme

3 STRAIN MONITORING DURING AUTOCLAVE CURE

Within this paragraph, the strain filed measured by embedded sensors is reported. Figure 2 reports thermo-mechanical strain evolution experienced by first FBG array (hereinafter-named FBG1) during all the manufacturing steps, on the right axis of the picture the composite temperature profile is drawn. This array have been arranged within the composite along 45 degrees angled plies.

In the first manufacturing stage (first cure step, heating-dwell-cooling) an overshoot in

deformation is clearly visible related to the cure of the polymer matrix while in further stages it disappears, since the material is fully cured. It is worth to notice that the sensors have been mounted at same radius and different position along cylinder axis; experimental data confirm the symmetric behaviour of the system: as matter of the facts, figures show the black and red curves overlapping each other with very good approximation



Figure 2: FBG1 array strain history over four manufacturing steps

During later manufacturing stages, at the end of the dwell time, the load applied by external additional material change the deformation filed experienced by sensor leading to horn-like strain profiles.



Figure 4: FBG4 array strain history over four manufacturing steps

Figure 4 reports deformation field experienced by array FBG4. This array has been placed within the composite during second manufacturing stage along 0° plies (fibres are arranged along cylinder axis).



Figure 5: strain history over four manufacturing steps for arrays installed in stage 2 and further

Figure 5 reports deformation histories experienced during further manufacturing stages. Pictures on the left side are related to arrays oriented on the 45° direction while right side refers to array arranged along cylinder axis. Each row of the figure reports data from stages 2, 3,4 respectively.

4 RESIDUAL STRAINS BUILT-UP MODELLING

4.1 Thermo-chemical model

Thermoset resins transform from a liquid state to a solid state at elevated temperatures, their cure progress should be described by empirical rate equations. Latter approach derives a

phenomenological mathematical description of curing process from experimental assessment on DSC (differential scanning calorimetry). The approach assumes the exothermic heat flow proportional to the degree of conversion. Different expressions have been proposed to model the rate of reaction [7-8], within this work the model used to fit the experimental data in both dynamic and isothermal conditions is the following:

$$\frac{d\alpha}{dt} = k_1 \left(1 - \alpha\right)^{r_1} + k_2 \left(1 - \alpha\right)^{r_2} \alpha \tag{5}$$

While the resin cure, three different morphological state will be passed, the resin converts from liquid to rubbery state and finally to solid state. Such different resin state are not discontinuity in real system, but they are usually modelled as discontinuous points function of glass transition temperature. The Di Benedetto's equation, is commonly used for computation purpose in the following form [9]:

$$T_{g}(\alpha) = \frac{(1-\alpha) \cdot T_{g0} + \lambda \cdot \alpha \cdot T_{g\infty}}{(1-\alpha) + \lambda \cdot \alpha}$$
(6)

Where T_{g0} and $T_{g\infty}$ are the transition temperature for the uncured system and for the fully cured system respectively, λ is a fitting parameter. The table below reports fitting parameter for the epoxy system considered within this work

$LogA_1$	E_1	r ₁	lgK _{ldiff}	logA ₂	E ₂	r ₂	a2	logK _{2diff}	C_1	C ₂
1/sec	kJ/mol		1/sec	1/sec	kJ/mol			1/sec		K
13.7	124.8	86.3	-17.3	8.6	80.2	2.2	1.1	95.1	84.1	143.4

4.2 Viscoelastic constitutive equations

The stress relaxation function for thermo-rheological simple materials should be modelled by a finite exponential series as [10]:

$$Q_{ij}(\alpha,\xi) = Q_{ij}^{\infty} + \left(Q_{ij}^{\infty} - Q_{ij}^{0}\right) \sum_{m=1}^{N} w_m \exp\left[-\frac{\xi(\alpha,T)}{\tau_m(\alpha)}\right]$$
(7)

The experimental data collected at different temperatures can be "shifted" relative to the frequency of loading, so that the various curves can be aligned to form a single master curve. The master curve can be constructed using an arbitrary selected reference temperature ($T_{ref} = 100^{\circ}C$) and shifted congruently.

Unidirectional tape employed have been experimentally investigated to determinate the 5 independent functions $Q_{11}(t)$, $Q_{22}(t)$, $Q_{12}(t)$, $Q_{23}(t)$, $Q_{44}(t)$ of the relaxation matrix in the constitutive stress strain material relation. Independent stiffness relaxation functions have been evaluated by considering their expression as function of independent engineering constants for transversely isotropic materials:

$$Q_{11}(\omega) \cong E_{11}(\omega); \ Q_{22}(\omega) \cong E_{22}(\omega) \left[\frac{1}{(1-v_{23}^2)} \right]; \ Q_{12}(\omega) \cong E_{22}(\omega) \left[\frac{v_{12}}{(1-v_{23})} \right]; Q_{22}(\omega) \cong E_{22}(\omega) \left[\frac{v_{23}}{(1-v_{23}^2)} \right]; \ Q_{44}(\omega) \cong G_{12}(\omega);$$
(8)

Once master curve is available according to procedure described, the relaxation moduli

and relaxation times should be evaluated by curve fitting procedure: the problem formulation for the evaluation of the master curve parameters is reported.

$$E(\omega) = E_0 + E_{\infty} - E_0 \cdot \sum_{i=1}^{10} w_i \left(\frac{\omega \tau_i^2}{1 + \omega \tau_i^2} \right)$$

$$\sum_{i=1}^{10} w_i = 1 , \ 0 \le w_i \le 1$$
(9)

4.3 Problem reduction – Finite elements formulation

Due to symmetry assessments, all derivatives with respect to the angular coordinate ϑ vanish. The equilibrium equations have been solved through the Finite Element Method. The discretization along the cylinder thickness is dependent by the stacking sequence and by the number of elements for each layer.

4.3.1 Heat transfer module

Within the autoclave, environment transfer energy to composite cylinder by convection. Energy equilibrium for axisymmetric problem, considering the exothermic reaction of the resin, are:

$$c_{p}\rho\frac{\partial T}{\partial t} = k\frac{\partial^{2}T}{\partial r^{2}} + k\frac{1}{r}\frac{\partial T}{\partial r} + \rho_{r}H_{r} - v_{f} \frac{\partial \chi}{\partial t}$$

$$\frac{\partial \chi}{\partial t} = f T, \chi$$
(11)

Where T is the temperature, χ is the degree of cure, r is the radius, t is the time, ρ is the composite density, ρ_r the resin density, k the radial conductivity, c_p the composite thermal capacity, H_r the cure Enthalpy, v_f the volumetric fibre fraction.

Boundary condition between composite and air on the external surface and between mould and air on the internal tool surface, have been imposed according to the following equations:

$$J_{ri} = h_{int} \begin{bmatrix} T_a & t & -T_i \end{bmatrix}$$

$$J_{re} = h_{ext} \begin{bmatrix} T_e - T_a & t \end{bmatrix}$$
(10)

Where J_{ri} is the internal radial heat flow, J_{re} is the external radial heat flow, h_{int} is the internal convective heat exchange coefficient, h_{ext} is the external convective heat exchange coefficient, $T_a(t)$ is the air temperature in the chamber, T_i is the temperature at the internal surface of the tool, T_e is the temperature at the external surface of the composite.



Figure 6: Thick composite cylinder in autoclave chamber under environmental conditions.

4.3.2 Mechanical module

In cylindrical coordinates, for axisymmetric problems the mechanical equilibrium equations become:

$$\frac{\partial \sigma_r}{\partial r} + \frac{\partial \tau_{rz}}{\partial z} + \frac{\sigma_r - \sigma_g}{r} = 0$$

$$\frac{\partial \tau_{rg}}{\partial r} + \frac{\partial \tau_{gz}}{\partial z} + 2\frac{\tau_{gz}}{r} = 0$$

$$\frac{\partial \tau_{rz}}{\partial r} + \frac{\partial \sigma_z}{\partial z} + \frac{\tau_{rz}}{r} = 0$$
(12)

Boundary conditions require that external surface is in equilibrium with the environmental pressure, in addition the external surfaces are free of transverse stresses:

$$\sigma_r R_{ext}, t = -p t ; \tau_{rg} R_{ext}, t = 0; \tau_{rz} R_{ext}, t = 0;$$
(13)

5 DISCUSSION

Longitudinal strains changes during the cure cycle are almost negligible (figure 5), it is worth to notice that strain are mainly negative since coefficient of thermal expansion of carbon fibre is slightly negative. Even in that case, the chemical shrinkage at cure onset is visible as a change in the strain peak before the temperature dwell.

Figure 7 reports the deformation history of array FBG1 over all steps on the same temporal axis. In the picture, the curves reproducing the strain level during isothermal phase (black curve) and residual strain level at the end of the step (green curve) have been reported. The strain level at plateau seems to be constant over all steps while residual strain dramatically increase when additional material increase in further manufacturing steps.



Figure 7: FBG1 array strain history over four manufacturing steps

Figure 5-a reports the strain profile in the case of array FBG3. Pictures reports the residual strain rise up during step2, step3 and step4. This array has been installed within second manufacturing stage; the overshoot visible at the installation stage refers to the chemical shrinkage due to resin cure. In further steps, overshoot following heating ramp are smaller, a rationale for such behaviour should be small residual cure in third and fourth stages.



Figure 8: Comparison between experimental data (dashed lines) and numerical (solid lines)

Comparison between numerical prediction and experimental results is presented in the figure 8), curves of the same colour refer to same radial position. Numerical predictions showed fair agreement with experimental data over all the manufacturing procedure the absolute differences are generally lower than $150\mu\epsilon$.

The differences between numerical results and experimental data are related to sensors arranged 45° direction respect to the cylinder axis. Experimental data show higher residual compression at the end of the cooling stage if compared with the numerical predictions.

Higher compression state in cooling stage is consistent with the hypothesis of "more hoop" configuration of sensors and/or the wrapped tape. At the warming up of the cylinder, the nominal ply orientation (45°) should be varied toward a more "hoop" configuration due to the thermal expansion of the tool. This reconfiguration is consistent with the fluid like status of the matrix before gelation. In addition, such discrepancy of the angle ply in respect with

nominal value is also consistent with anomalies during the wrapping phase.

Figure 9 show the comparison between numerical and experimental strain histories considering within the model a deviation to a more hoop condition of 45° ply.



Figure 9: Comparison with experimental data

6 CONCLUSIONS

In this work, a cure monitoring system using FBG sensors was devised to monitor the cure cycle of a thick composite component with the purpose of reducing the thermal residual stresses. The thermos-viscoelastic residual stresses residual stresses within a polymer composites cylinder made by multiple manufacturing stages were experimentally investigated and numerically modelled.

The strain evolution during manufacturing steps showed that the addition of composite plies in multiple manufacturing step lead to the rise of a compressive strain state on the cylinder.

The numerical procedure proposed was able to reproduce experimental data, but some discrepancies were detected considering sensor arranged on tapes ringing around the cylinder. The model was corrected by considering a possible mismatch to the nominal orientation of such plies due to a possible rearrangement before gelation driven by tool expansion.

ACKNOWLEDGEMENT

This activity has been performed within the seventh framework program, grant "SCYPRI-Smart Cylinders for Flexographic Printing Industry". The Authors would like to thanks Reglass staff who actively contribute to in-autoclave experimentations.

REFERENCES

- [1] Fonseca G.E., Dubè M.A., Penlidis A., A Critical Overview of sensors for Monitoring Polymerizations. Macromolecular Reaction Engineering, Vol 3, pp327-373, 2009.
- [2] Konstantopoulos S., Fauster E., Schledjewski R., Monitoring the production of FRP composites: A review of in-line sensing methods. Express Polymer Letters, Vol. 8, pp. 823-840, 2014.
- [3] V. Antonucci, M. Giordano, A. Cusano, J. Nasser, L. Nicolais, Real time monitoring of cure and gelification of a thermoset matrix, Composites Science and Technology, Vol. 66, 2008.
- [4] Fernando G.F., Fibre optic sensor systems for monitoring composite structures. Reinforced Plastics, Vol. 49, pp. 41-50, 2005.
- [5] D. Karalekas, J. Cugnoni, J. Botsis, Monitoring of process induced strains in a single fibre composite using FBG sensor: A methodological study, Composites Part A: Applied Science and Manufacturing, Volume 39, Issue 7, July 2008, Pages 1118-1127, ISSN 1359-835X, http://dx.doi.org/10.1016/j.compositesa.2008.04.010.
- [6] Antonucci V., Giordano M., Imparato S.I., Nicolais L., Autoclave Manufacturing of Thick Composites, Polymer Composites, Vol. 23, Issue 5, 2002.
- [7] Hale A., Macosko C.W., Bair E.H., Glass transition temperature as a function of conversion in thermosetting polymers, Macromolecules, Vol 24 (9), pp. 2610-2621, 1991.
- [8] Luo Z., Wei L., Li W., Liu F., Zhao T., Isothermal differential scanning calorimetry study of the cure kinetics of a novel aromatic maleimide with an acetylene terminal, Journal of Applied Polymer Science, Vol 109, pp. 1097-4628, 2006.
- [9] Di Benedetto A. T., Prediction of the glass transition temperature of polymers: A model based on the principle of corresponding states, Journal of Polymer Science Part B: Polymer Physics, Vol 25, pp. 1949-1969,1987.
- [10] Kim Y.K., White S.R., Stress relaxation behaviour of 3501-6 epoxy resin during cure, Polymer Engineering & Science, Vol.36, pp. 2852-2862, 1996.
- [11] Maffezzoli, A Grieco, A., Optimization of Parts Placement in Autoclave Processing of Composites Journal of Applied Composite Materials, Vol. 20 (3), 2012.